
Measurements and Models of Cytoskeletal Rheology

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MIT

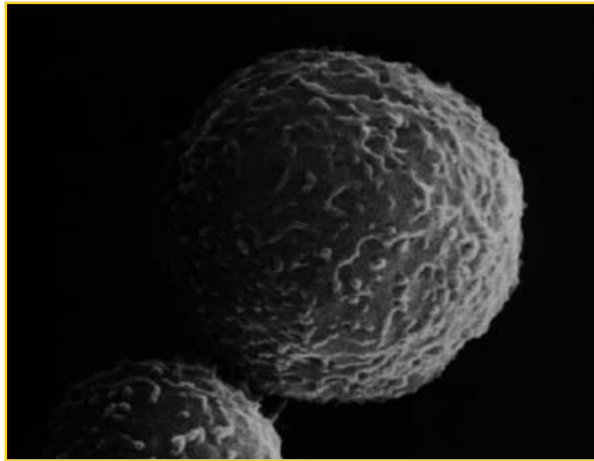
Department of Mechanical Engineering

Department of Biological Engineering

Acknowledgements: TaeYoon Kim, Hyunsuk Lee, Belinda Yap,
Jeff Hsu, Jorge Ferrer, Seok Chung

The Neutrophil -- naturally-occurring deformations

SEM



(micrograph courtesy of C. Dong and R. Skalak)

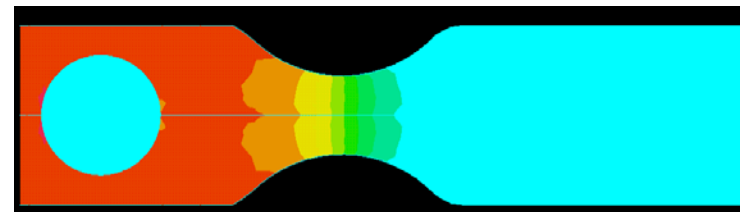
- spherical
- mean diameter = $6.8 \mu\text{m}$

Cross-section



(micrograph courtesy of C. Dong and R. Skalak)

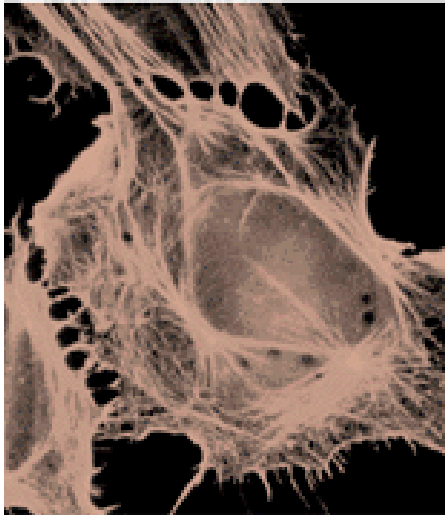
- multi-lobed nucleus
- granules/cytoskeleton
- actin-rich cortical region



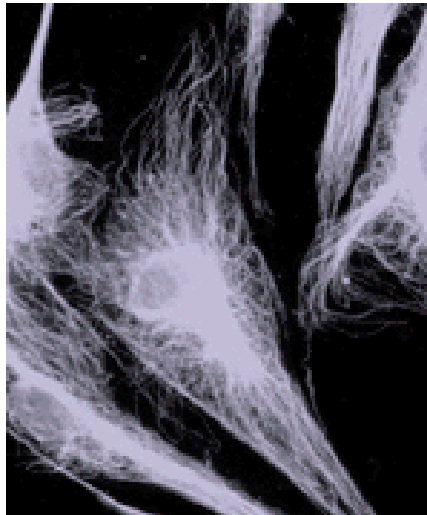
Bathe et al., 2002

Cytoskeletal composition/structure

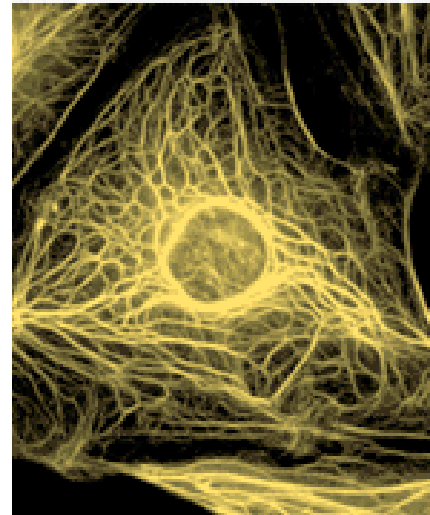
MICROFILAMENTS



MICROTUBULES

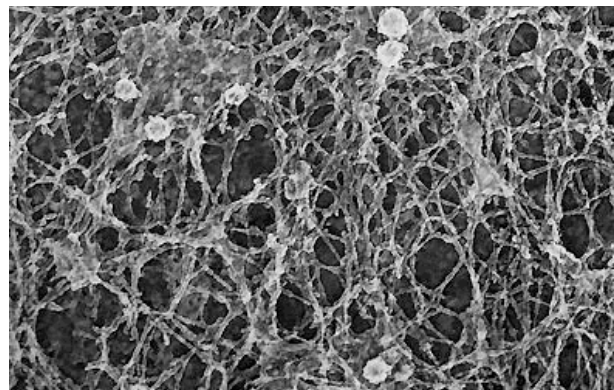


INTERMEDIATE FILAMENTS



Ingber, Scientific American

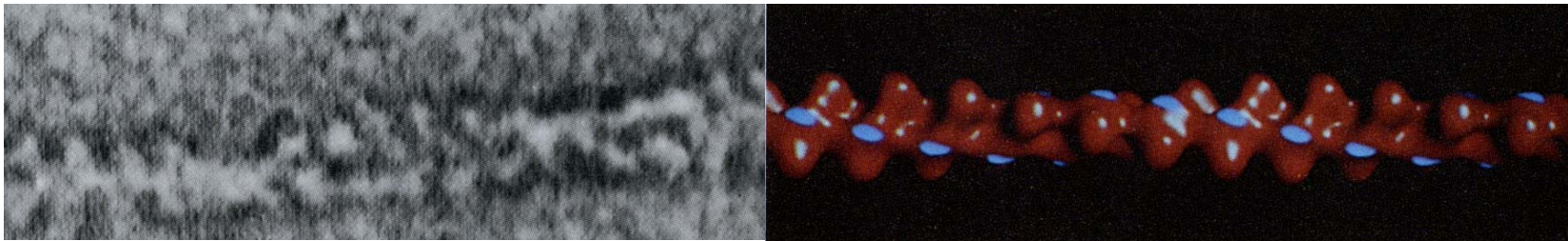
Cells have a multi-component fibrous, gel-like matrix.



Hartwick,
<http://expmed.bwh.harvard.edu>

Actin -- One of the primary structural components of cells

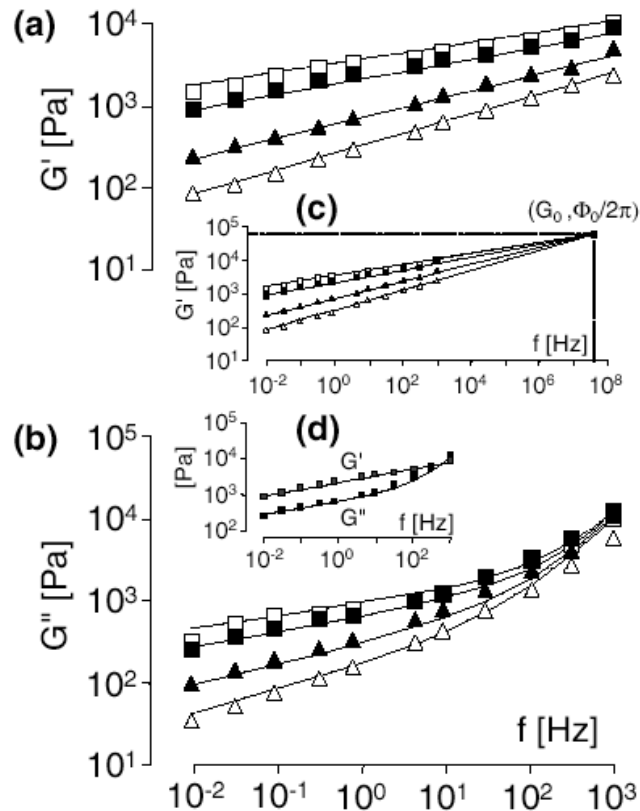
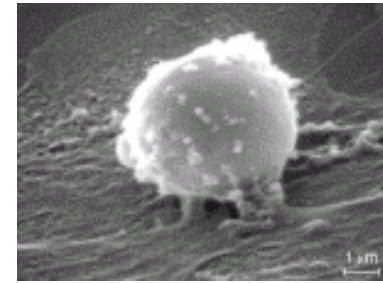
- 40kDa (375 AA residues)
- G-actin to F-actin : Nucleation (lag phase), elongation (growth phase), steady state (equilibrium phase)



Biochemistry, Wiley 1995

- Semi-flexible filament (neither flexible nor stiff) $l_p \approx l_c$
- Viscoelastic network
- Monomer size \ll mesh size $\leq l_p$
- Self assembly with polarity
- Binding to a variety of ABPs (Actin Binding Proteins)
- Actin polymerization plays a key role in fundamental cellular processes

Magnetic Twisting Cytometry (MTC)



$$G^* = G' + iG'' = \alpha \frac{\tilde{T}(t)}{\tilde{\delta}(t)}$$

α is a geometry-dependent prefactor determined from finite element analysis

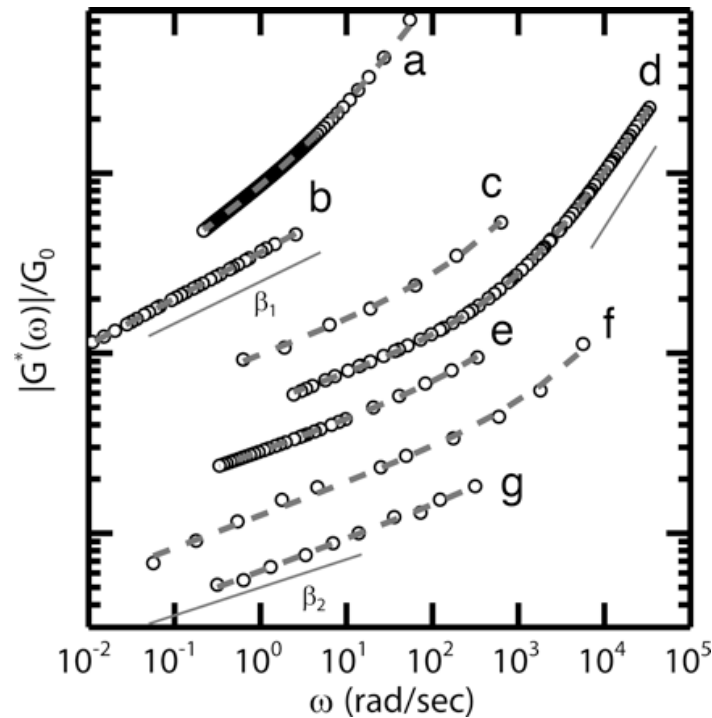
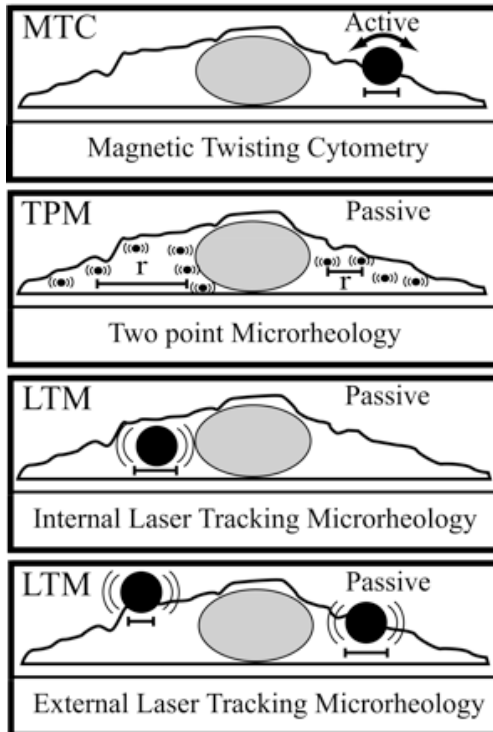
$$G^* = G_0 \left(\frac{\omega}{\omega_0} \right)^{x-1} (1 + i\bar{\eta}) \Gamma(2-x) \cos \left[\frac{\pi}{2}(x-1) \right] + i\omega\mu$$

Γ is the Gamma-function; G_0 , Φ_0 and x are adjustable parameters

Consistent with models for soft, glassy materials

Cells exhibit a power-law rheology, reminiscent of a soft, glassy material.

Different behaviors for different types of measurement



Data appear to fall along two characteristic slopes, one for methods that emphasize cortical structure, and another for internal, cytoskeletal structure.

But values for G range from 10's of Pa to several kPa, depending on the method of measurement.

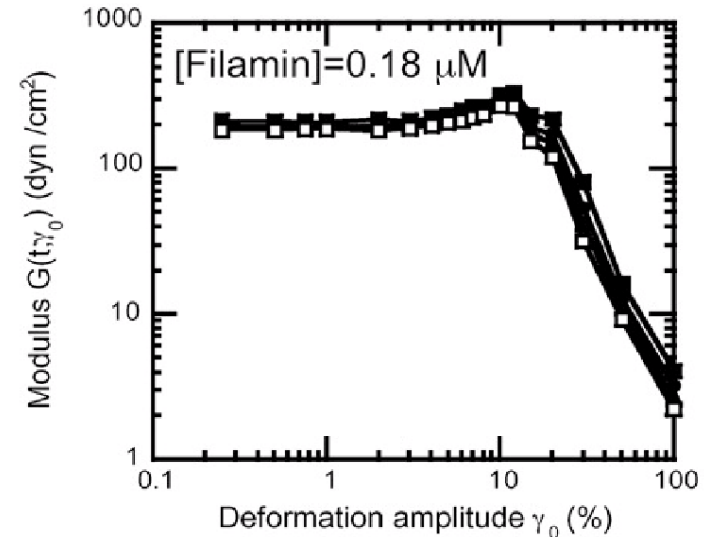
Hoffman, et al., PNAS, 2006

APS-DFD
November, 2006

Reconstituted actin gel rheology

Hypotheses for breakdown of the cytoskeleton due to mechanical deformation

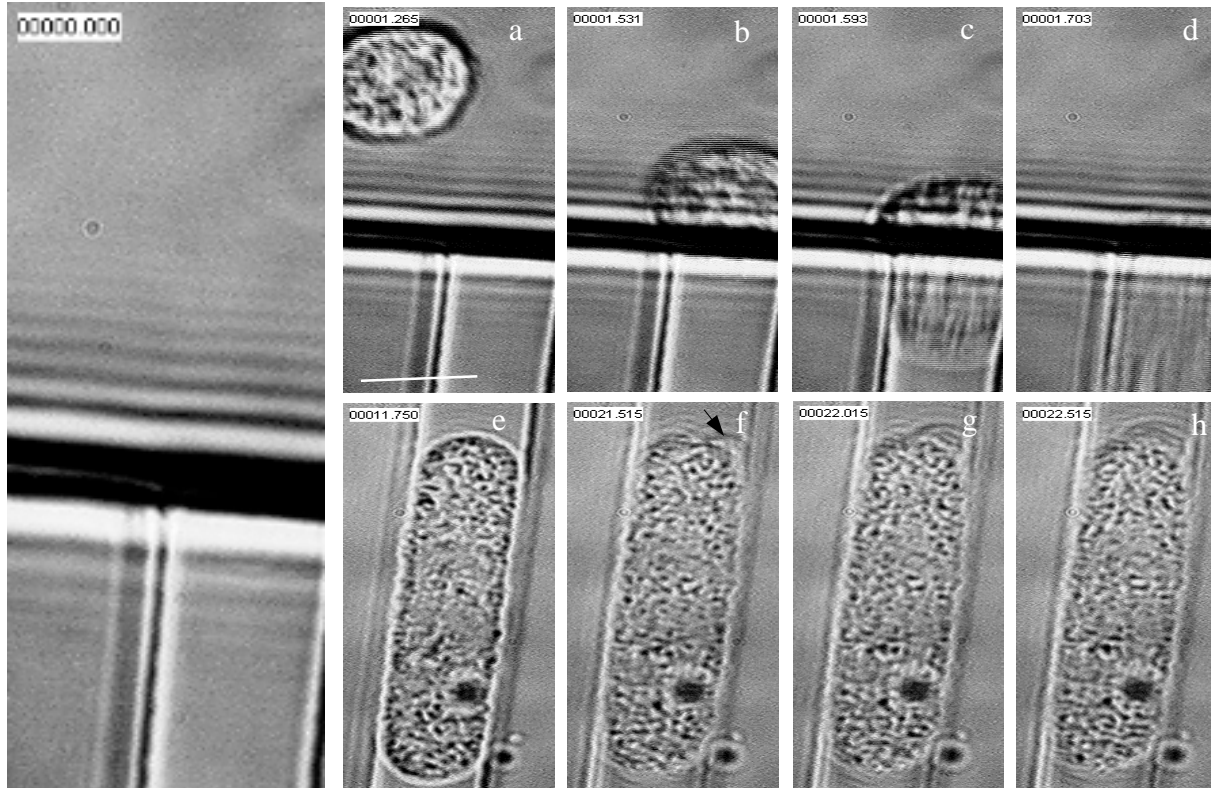
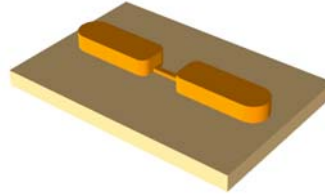
- 1) Sudden depolymerization of F-actin
- 2) Rupture of actin cross-links between F-actin filaments



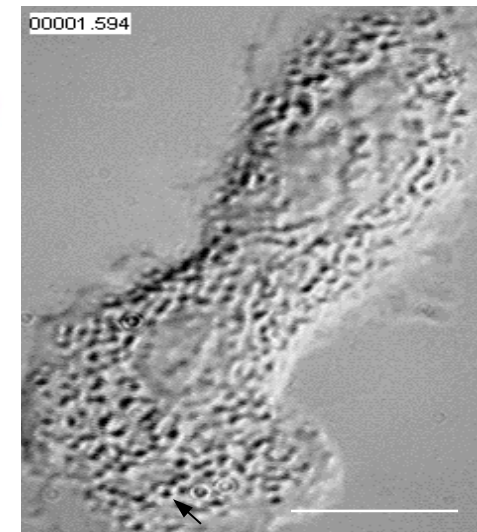
Tseng et. al. (2004) J Biol Chem; 279: 1819-1826

Incremental shear moduli exhibit strain hardening up to a point, followed by a catastrophic drop in stiffness.

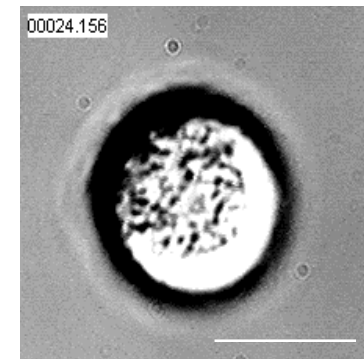
Cytoskeletal shear modulus computed from the Brownian fluctuations of granules



As a function of time after entering
channel, until protrusion

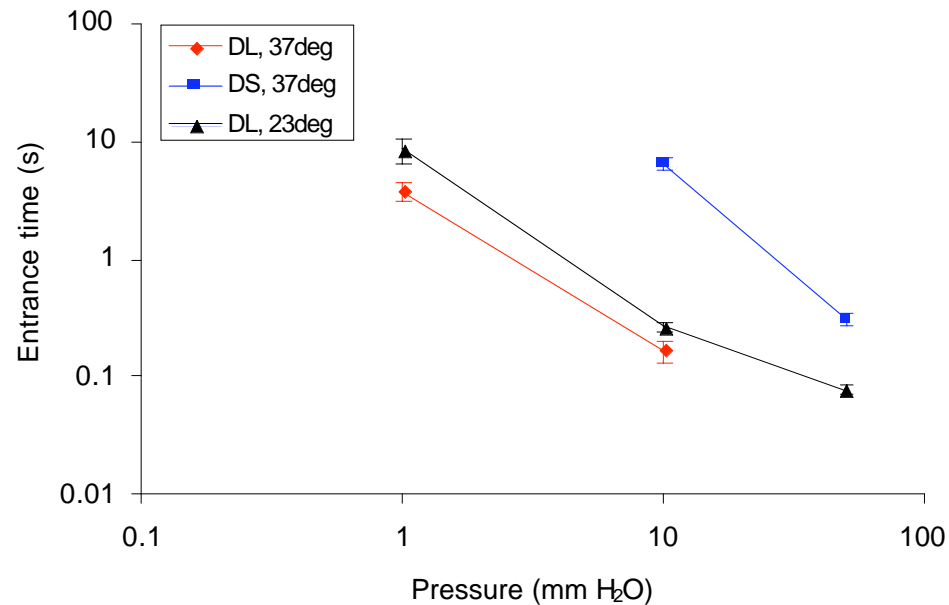


In adherent, migrating
neutrophils

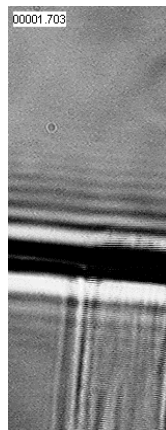
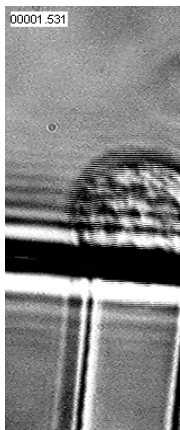


In non-adherent,
neutrophils

Entrance Time -- Passive Behavior



Entrance time appears primarily influenced by the magnitude of mechanical stimulation; less so by biological activity

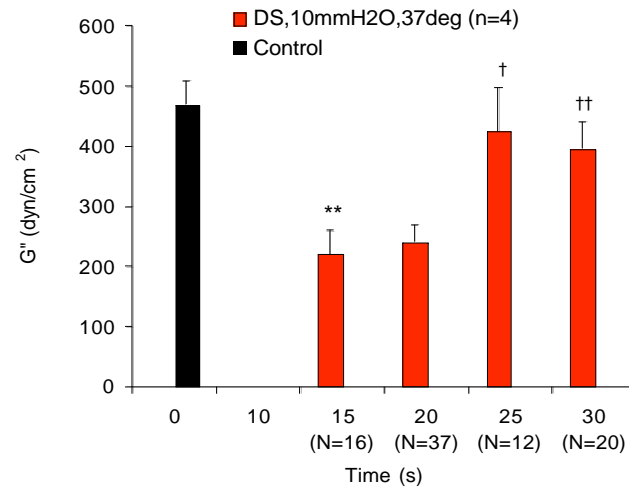
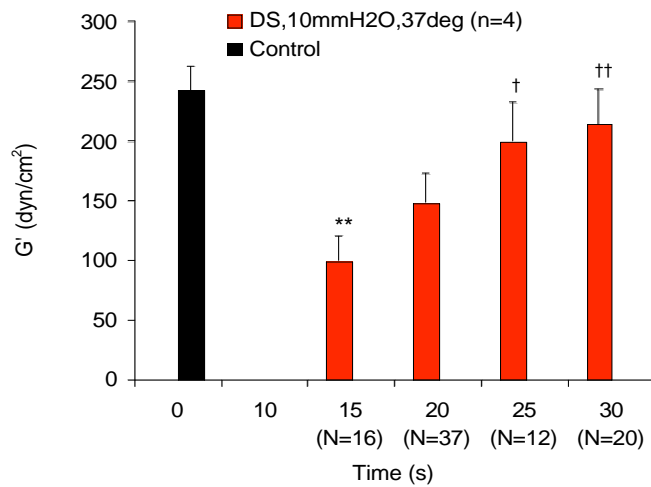
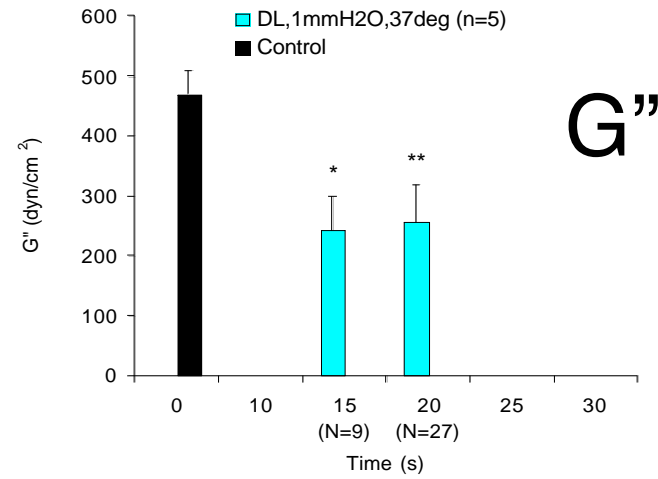
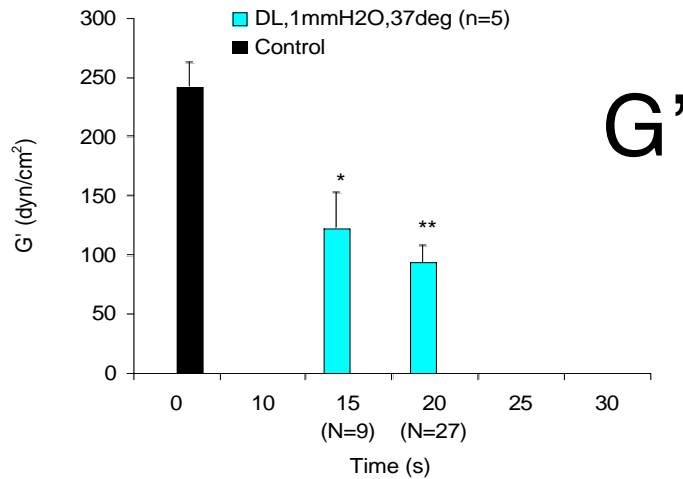


Entrance time : interval between leading edge touching channel entrance and trailing edge clearing channel entrance after deformation

To what extent can the cytoskeleton be treated as a cross-linked actin gel?

Yap & Kamm, APS, 2005

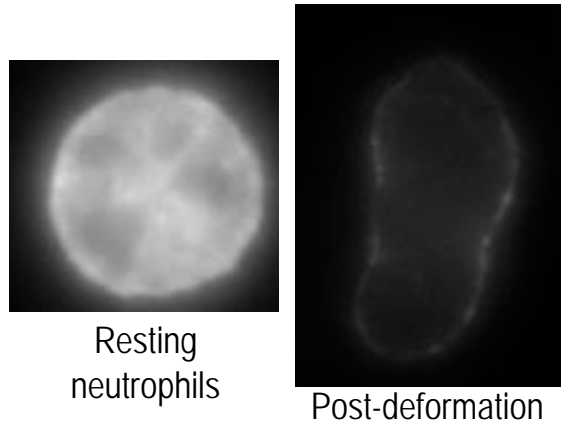
Viscoelastic moduli of neutrophils following deformation



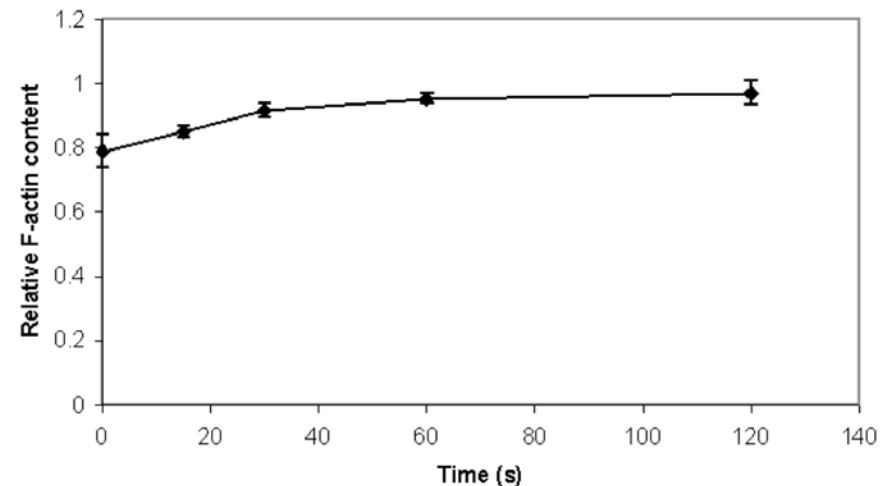
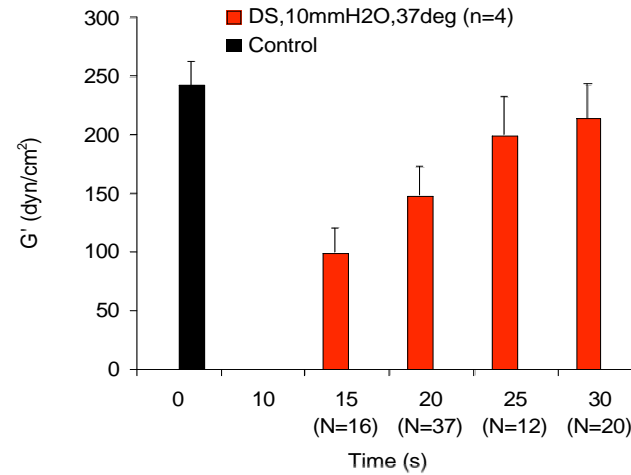
Moduli fall immediately upon deformation, and recover on a time scale of about one minute

Dynamic shear modulus and F-actin follow similar patterns

Storage modulus drops immediately, then recovers on a time scale of 30-60s.

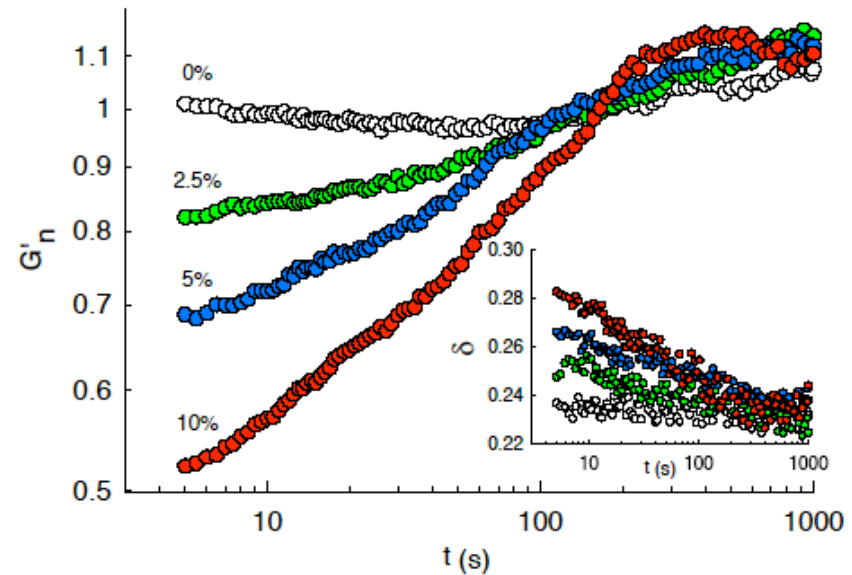
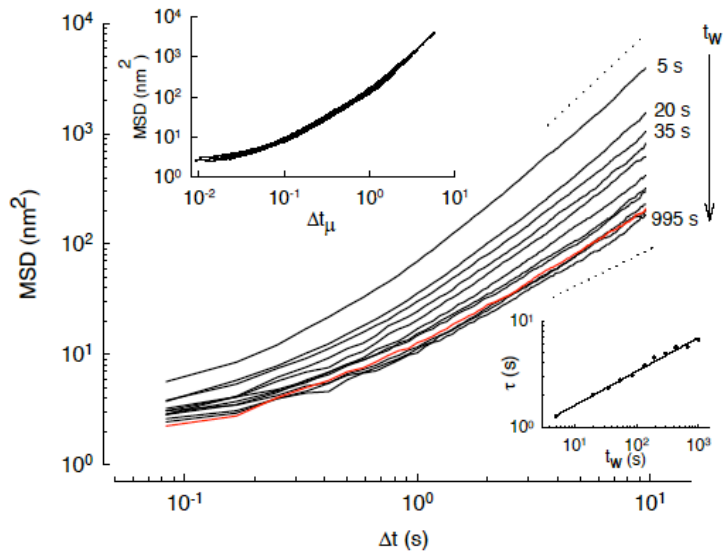
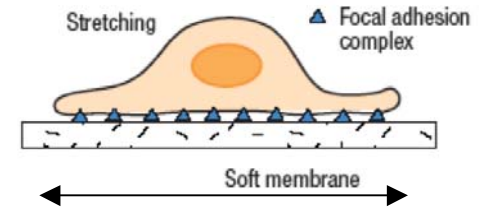


A similar time-course is observed for F-actin content following passage through pores.



Actin depolymerization appears to explain part, but perhaps not all, of the drop in modulus

Similar behavior -- abrupt fall, then recovery of stiffness -- has been observed in other cell types



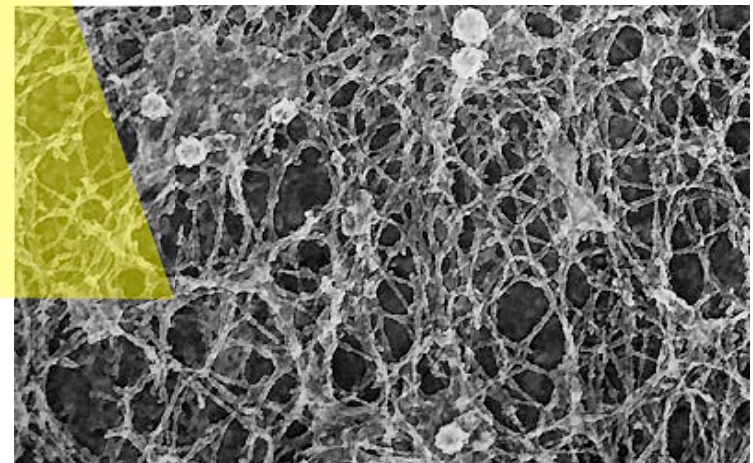
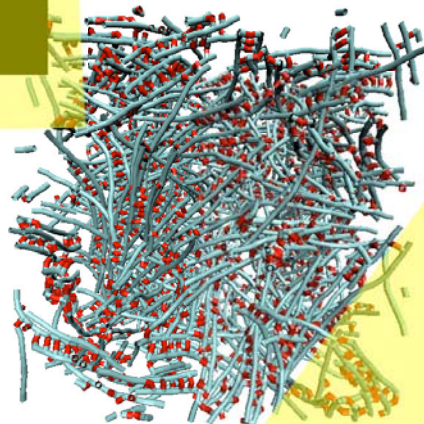
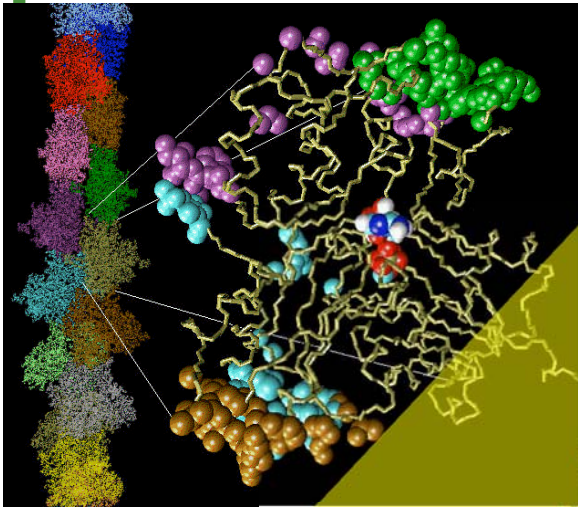
Universal physical responses to stretch in the living cell

10 November 2006

Xavier Trepat¹, Linhong Deng^{1,2}, Steven S. An^{1,3}, Daniel Navajas⁴, Daniel J. Tschumperlin¹, William T. Gerthoffer⁵, James P. Butler¹, and Jeffrey J. Fredberg¹

Cytoskeletal fluidization is a common attribute of cells

Cytoskeletal remodeling



Forces are transmitted via the CSK and actin cross-links

Bonds may rupture and reform, leading to remodeling

Signaling pathways might initiate depolymerization

Some outstanding questions

What is the physical basis for soft, glassy rheology?

How can strain stiffening in reconstituted actin gels be reconciled with fluidization in cells under abrupt strains?

Are the differences in behavior seen with different measurement methods real, and how can they be reconciled?

Are the different power law behaviors really due to cortex vs. cytoskeleton? If so, why?

Modeling Objectives

Simulate actin cytoskeletal growth, rheology, and force-induced changes in biochemical activity using molecular dynamics, Brownian dynamics, and continuum models.

Cytoskeleton Growth

- Investigate the effects and roles of the properties of actin cross-linking proteins (ACPs) on cytoskeletal structure

Microbead Rheology

- Estimate viscoelasticity of the actin cytoskeleton using microbead tracking rheology, both active and passive.
- Investigate the effect of ACPs on viscoelasticity

Relevant recent works: Storm, et al., Nature (London), 2005
Hoffman, et al., <http://arxiv.org/pdf/physics/0504051>

Cytoskeletal Modeling

Equation of motion: Langevin equation

$$m \frac{d^2 \vec{r}}{dt^2} = \sum_{i \neq j} \vec{f}_{ij} - \zeta \frac{d\vec{r}}{dt} + \vec{d}(t)$$

Drag computed (initially) ignoring hydrodynamic interactions

Integration Method: Euler forward (dropping inertia)

$$\frac{d\vec{r}^*}{dt^*} = \sum_{i \neq j} \vec{f}_{ij}^* + \vec{d}^*(t) \rightarrow d\vec{r}^* = \left[\sum_{i \neq j} \vec{f}_{ij}^* + \vec{d}^*(t) \right] dt^*$$

Interaction forces:

Truncated Lennard-Jones

$$U_{LJ}(r) = \begin{cases} 4\epsilon \left[\left(\frac{\sigma}{r} \right)^{12} - \left(\frac{\sigma}{r} \right)^6 \right] + \epsilon & r \leq 2^{1/6} \sigma \\ 0 & r > 2^{1/6} \sigma \end{cases}$$

Scalings:

Length scaled with actin radius, energy with $k_B T$

Bending stiffness

$$U_{Bend} = \frac{1}{2} \kappa_{Bend} (\theta - \theta_0)^2$$

Extensional stiffness

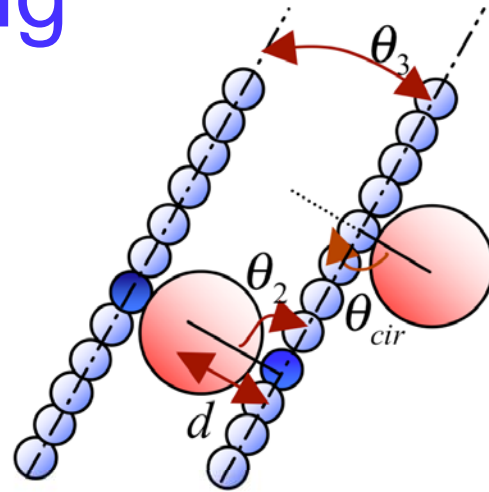
$$U_{Spring} = -\frac{1}{2} k_s (r - r_0)^2$$

Torsional stiffness

$$U_{Torsion} = \frac{1}{2} \kappa_{Torsion} (\theta - \theta_0)^2$$

$$t^* = \frac{t}{\sigma^2 \zeta / k_B T}$$

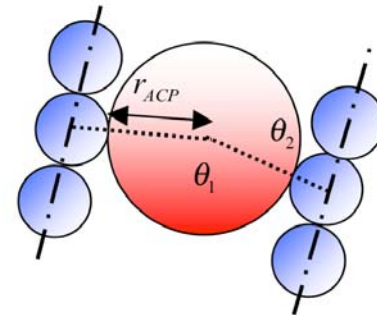
Cross-linking



Unbinding obeys Bell's equation and is force-dependent.

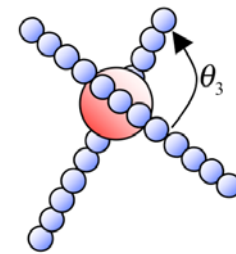
Bundling cross-linkers (e.g., fimbrin, fascin, scriuin, α -actinin)

$$r_{ACP} = 0.75, \theta_1 = \pi, \theta_2 = \frac{\pi}{2}, \theta_3 = 0$$



Network cross-linkers (e.g., filamin)

$$r_{ACP} = 1.5, \theta_1 = 1.158 \text{ (rad)}, \theta_2 = \frac{\pi}{2}, \theta_3 = \frac{\pi}{2}$$



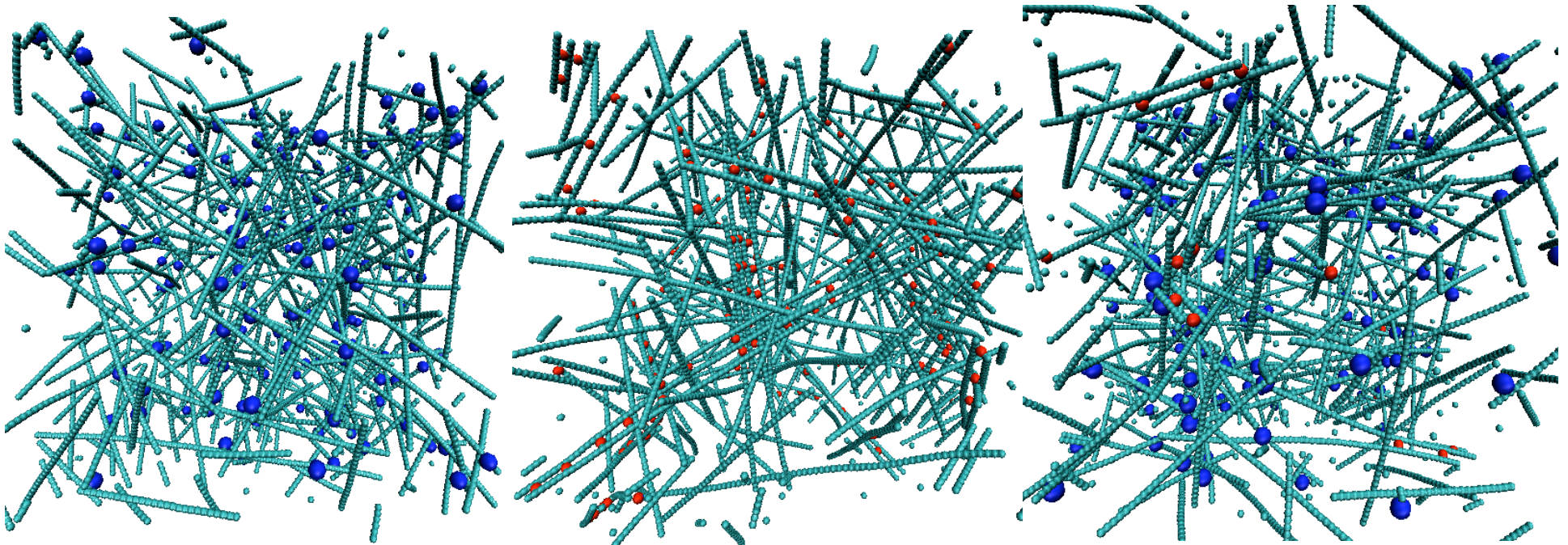
Comparison of cross-linked and bundled structures

(varied by changing the torsional stiffness and orientation of actin binding sites)

Cross-linked

Bundled

Mixed



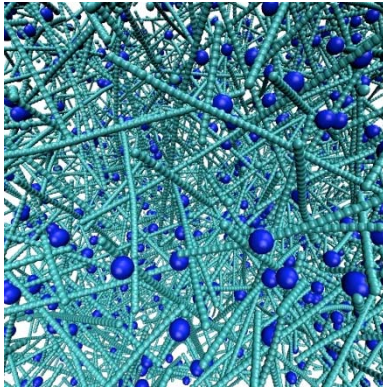
Monomers : 8,000, ACPs : 1,000

Vol. % = 0.82% (+ACP ~0.1%), $R=0.125$

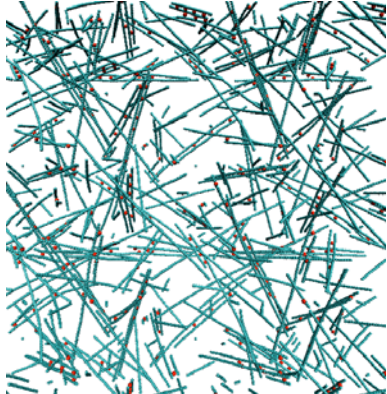
Conc = 150 μ M, Domain = 560 nm

Computed Network Structures

90° cross-links



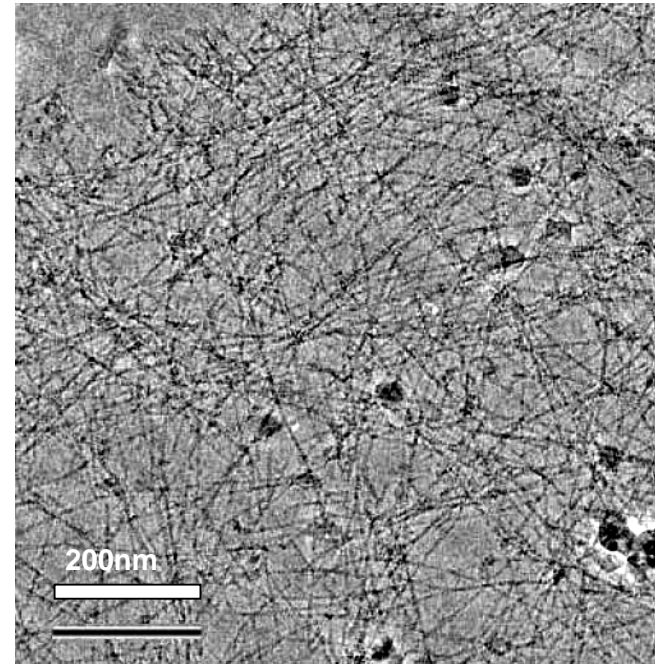
“Bundling” cross-links



Computed 3D images

“Confocal” images

1.12 μm x 1.12 μm

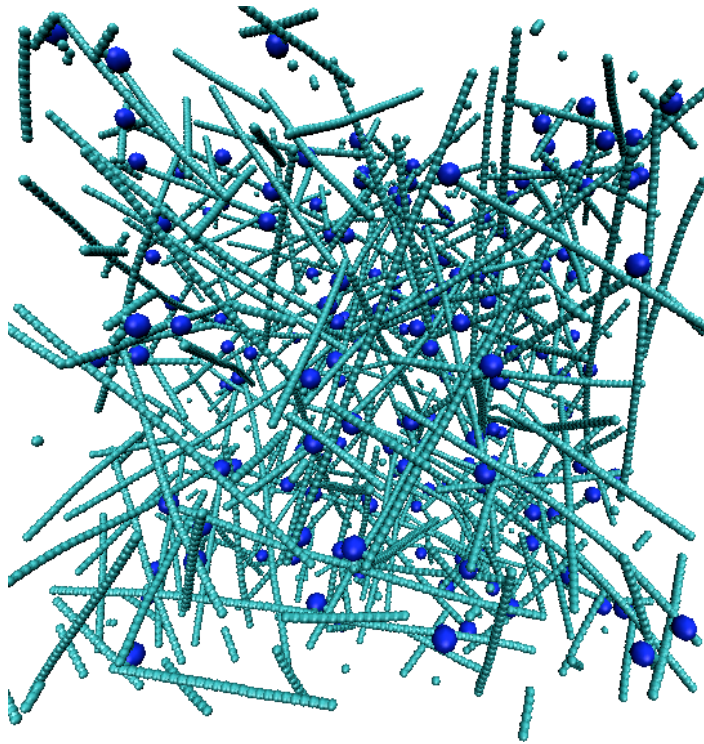


Visualisation of the actin cytoskeleton by cryo-electron microscopy

Guenter P. Resch¹, Kenneth N. Goldie², Angelika Krebs², Andreas Hoenger² and J. Victor Small^{1,*}

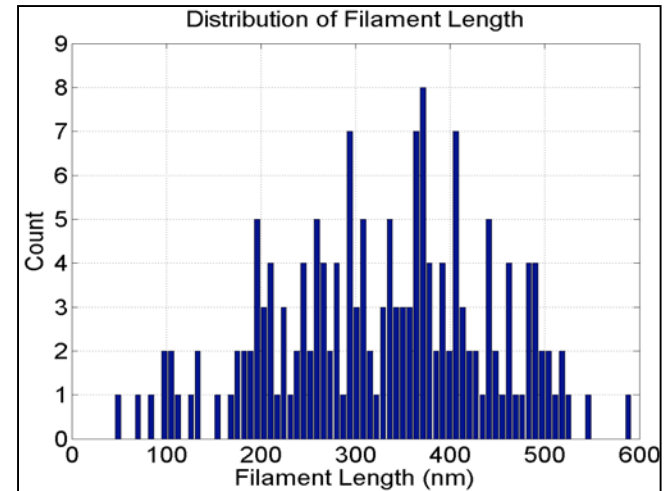
Journal of Cell Science 115 (9)

Analysis of the Polymerized Structure

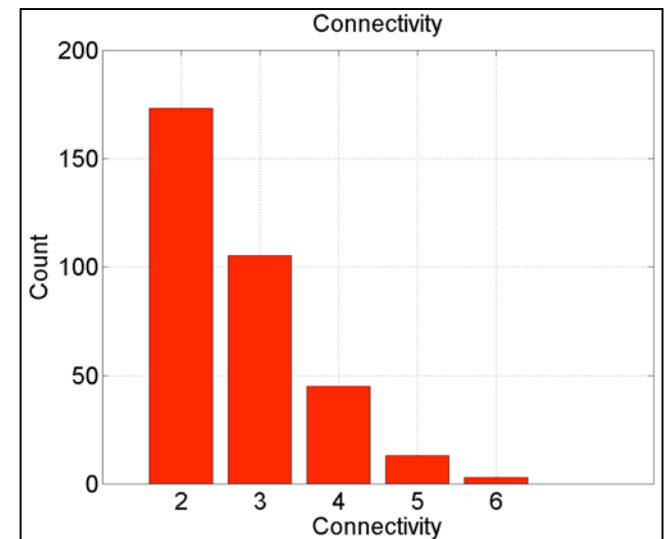


Monomers : 8000, ACPs : 173
Vol. % = 0.82% = 150 μ M
Nucleation rate = $1 \times 10^{-8} \text{ s}^{-1}$, Domain = 560 nm

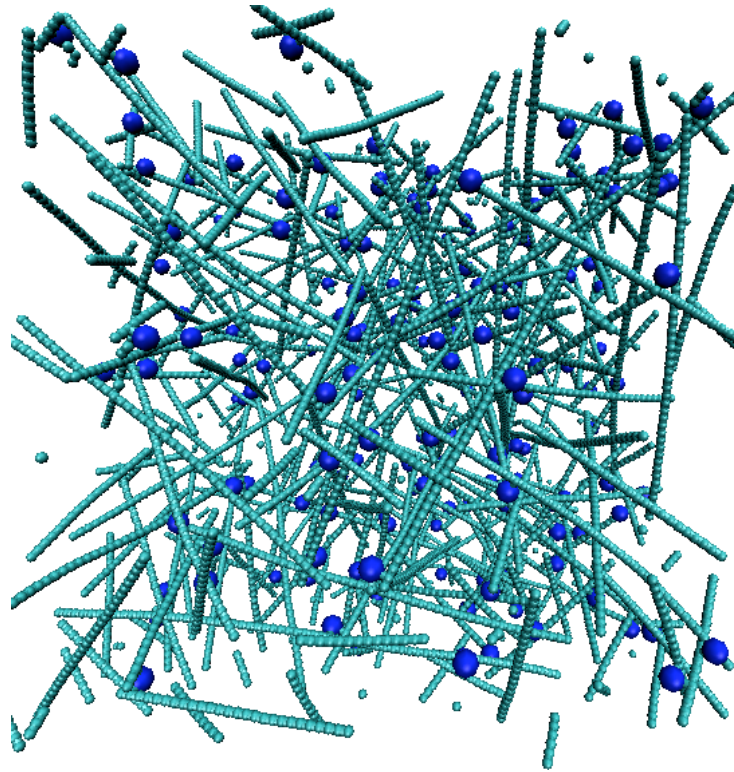
Filament length = total contour length
Connectivity = number of filaments attached
via cross-links



Avg = 328, S.D. = 112

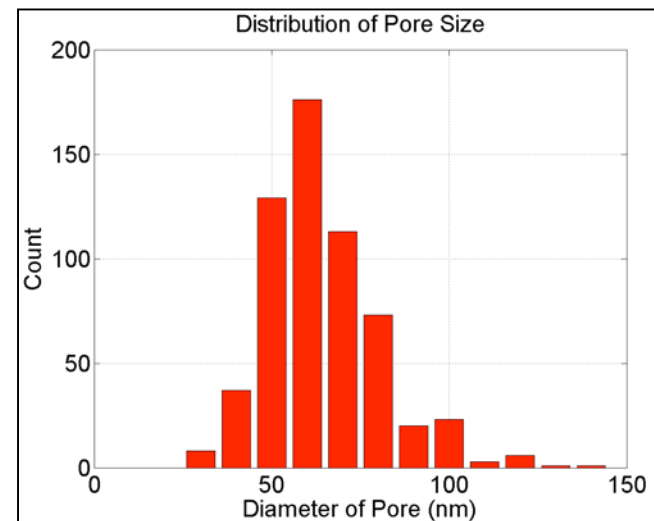
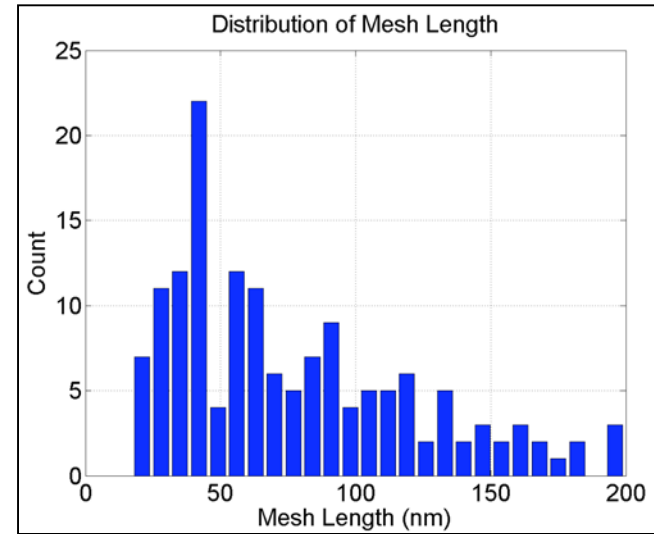


Analysis of the Polymerized Structure



Monomers : 8000, ACPs : 173
Vol. % = 0.82% = 150 μ M
Nucleation rate = $1 \times 10^{-8} \text{ s}^{-1}$, Domain = 560 nm

Mesh size = distance between cross-links
Pore size = max diameter of a sphere that can fit inside a pore



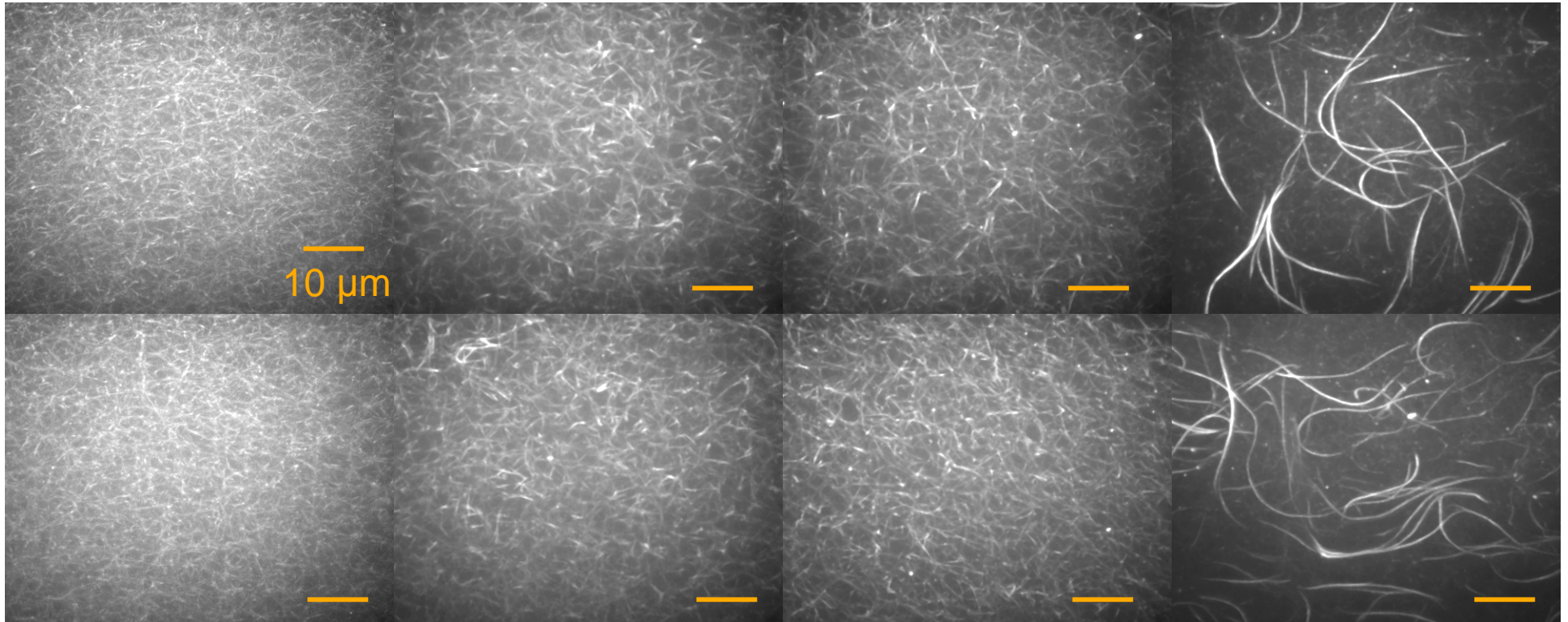
F-actin networks formed by α -actinin

$R_\alpha = 0.01$

$R_\alpha = 0.1$

$R_\alpha = 0.2$

$R_\alpha = 0.5$



Stress fibers form of increasing diameter as the amount of α -actinin is increased

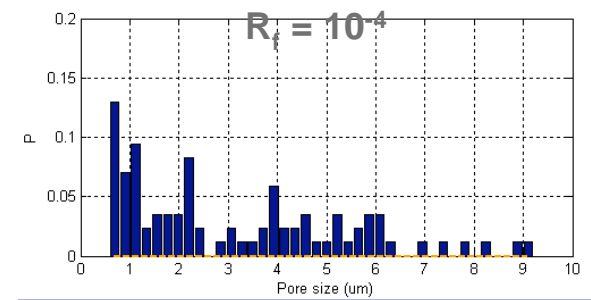
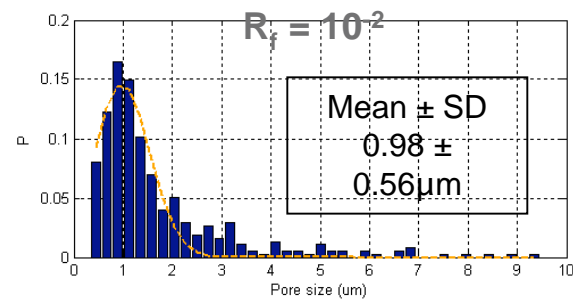
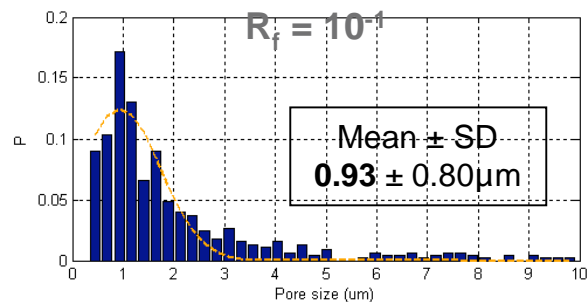
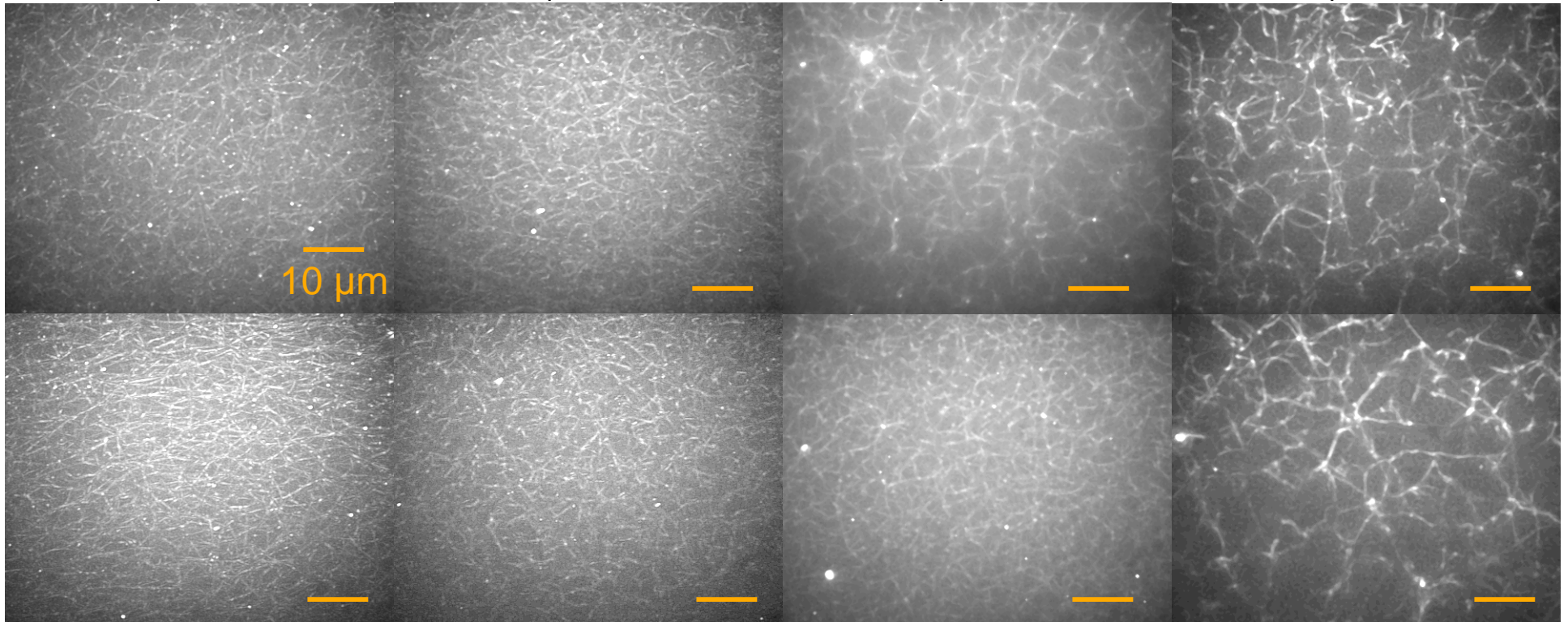
F-actin networks formed by filamin

$R_f = 10^{-1}$

$R_f = 10^{-2}$

$R_f = 10^{-3}$

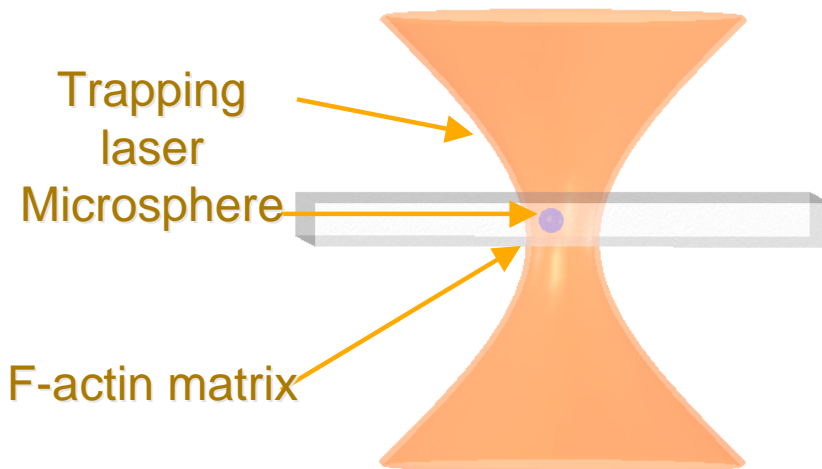
$R_f = 10^{-4}$



Passive microrheology

Monitor thermal fluctuations of embedded microspheres to estimate the frequency dependent complex modulus

High spatial and temporal resolution using quadrant photo detector



Thermally driven dynamics of the microsphere

Fluctuation-dissipation theorem

Kramers-Kronig relation

Generalized Stokes-Einstein relation

$$|G^*(\omega)| \approx \frac{k_B T}{\pi a \langle \Delta r^2(1/\omega) \rangle \Gamma[1 + \alpha(\omega)]}$$

$$G'(\omega) = |G^*(\omega)| \cos(\pi\alpha(\omega)/2)$$

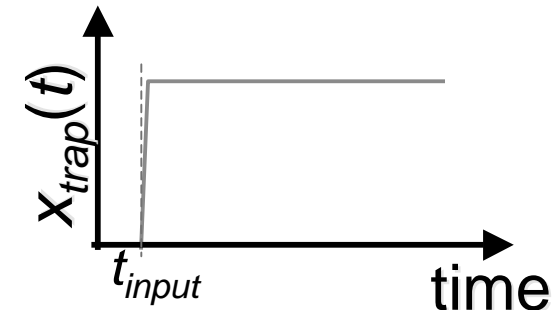
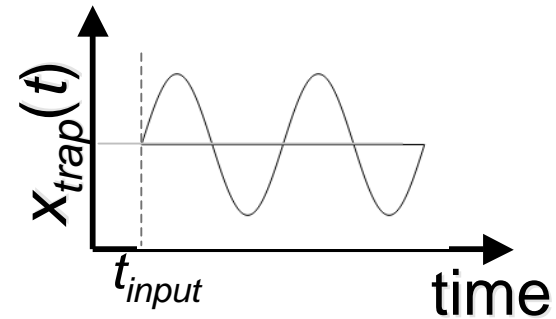
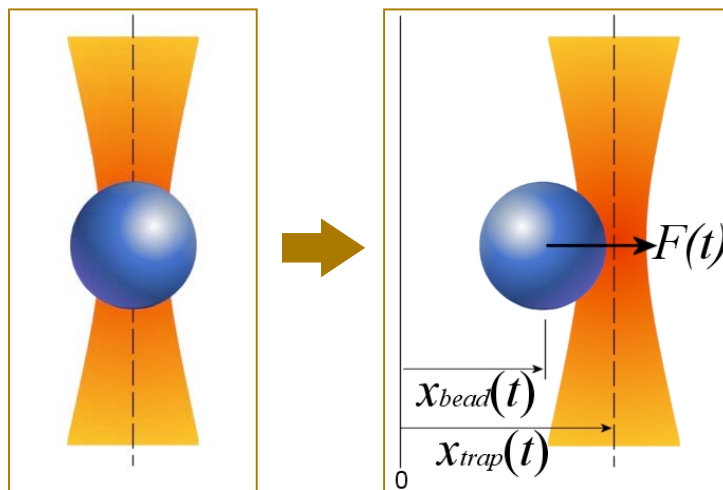
$$G''(\omega) = |G^*(\omega)| \sin(\pi\alpha(\omega)/2)$$

$$\alpha(\omega) \equiv \left. \frac{d \ln \langle \Delta r^2(t) \rangle}{d \ln t} \right|_{t=1/\omega}$$

Active microrheology

Apply a driving force to an embedded microsphere and monitor its trajectory

- Sinusoidal and stepwise

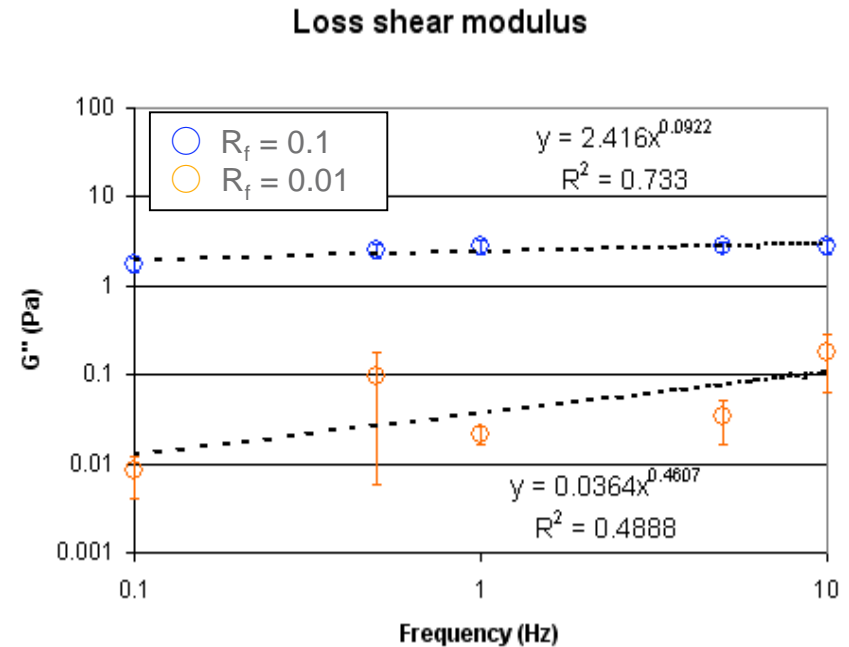
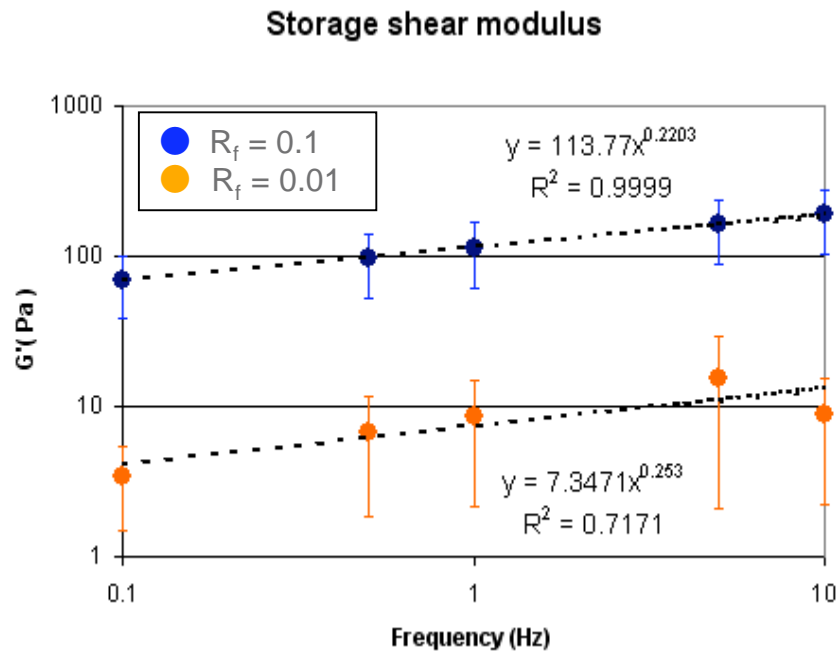


Bulk rheometer

- Parallel plate, AR2000, *TA Instruments*
- Gap: 120 μ m – 40mm flat plate
- Frequency sweep

Comparison at different filamin concentrations

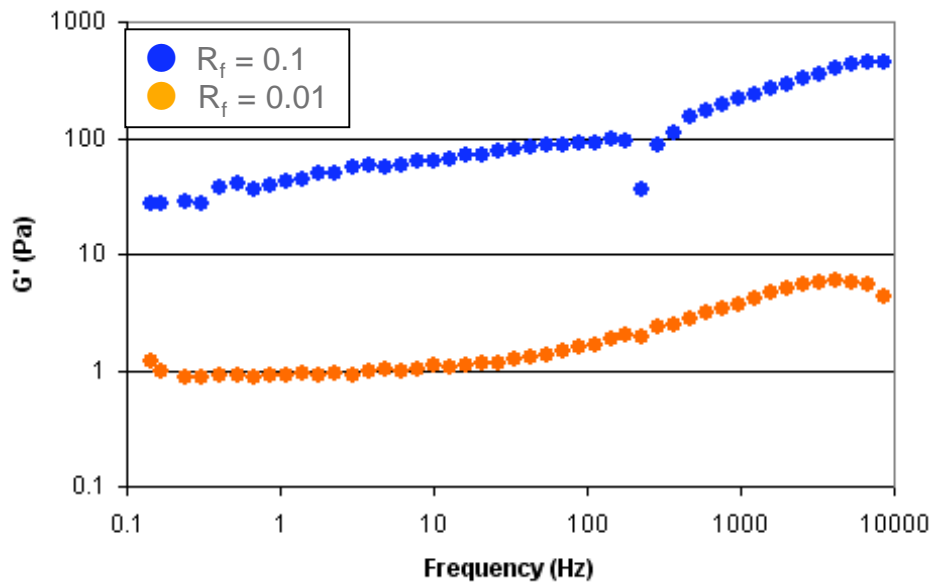
– Active sinusoidal forcing



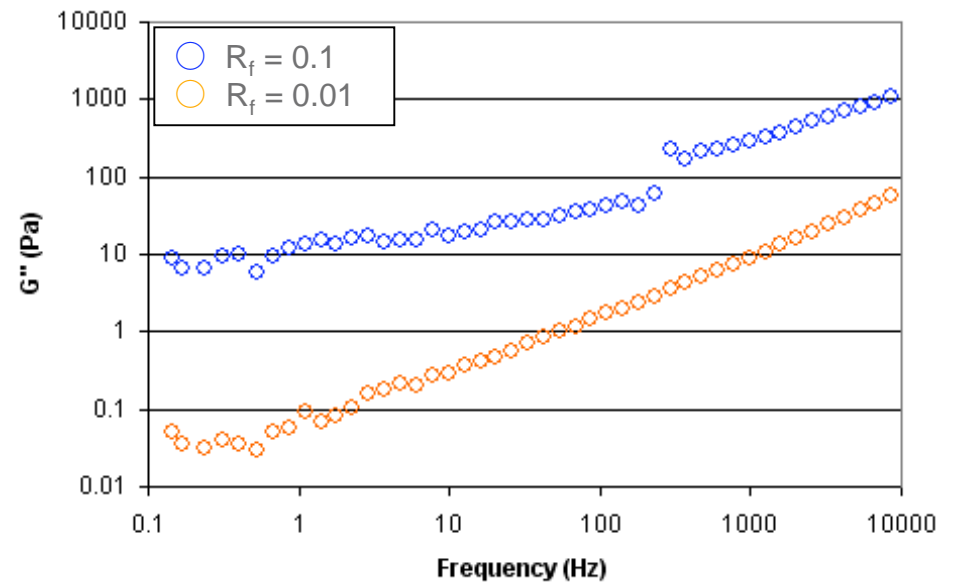
- One order magnitude difference of moduli as R_f increases 10 fold
- $G'_a \sim R_f^{0.2}$ for 10 μM actin, $R_f = 0.01$ and 0.1
- G'_a of $R_f = 0.1$ is comparable to cellular rheology

Comparison at different filamin concentrations – Passive measurement

Storage shear modulus

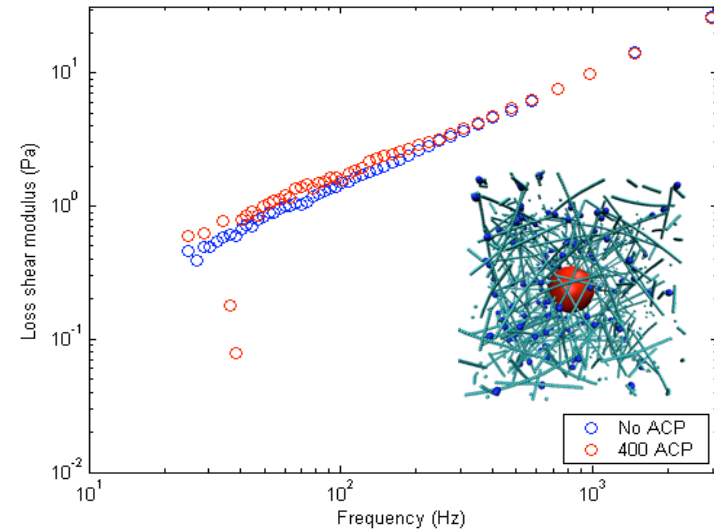
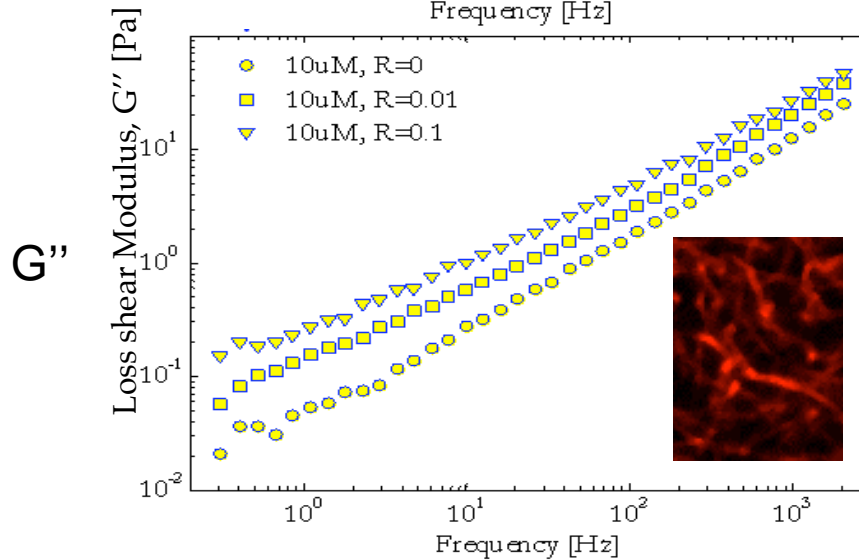
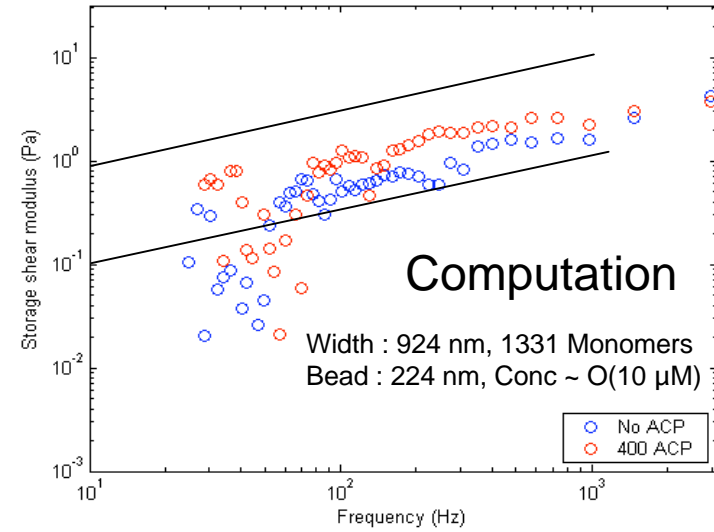
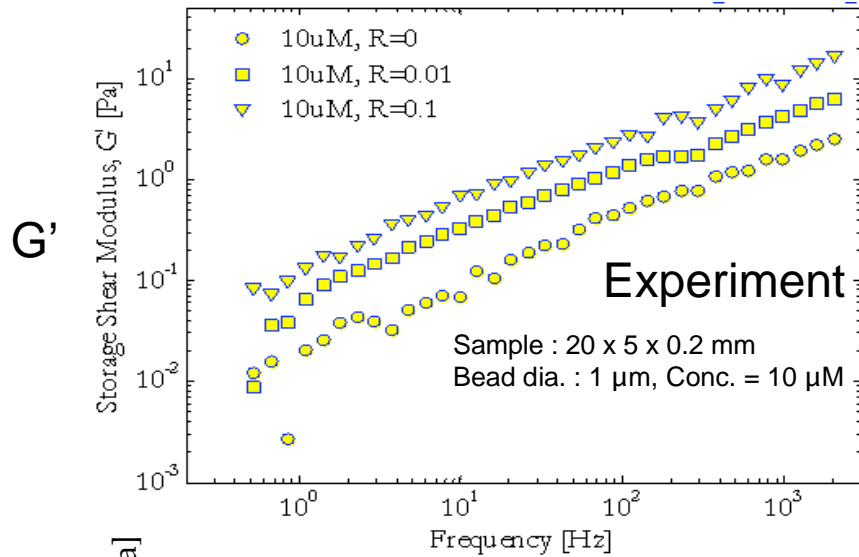


Loss shear modulus



- Longer plateau region in G'_p of $R_f = 0.01$
- Similar values of G'_p in low frequency region

Comparison of complex moduli



Summary and remaining questions

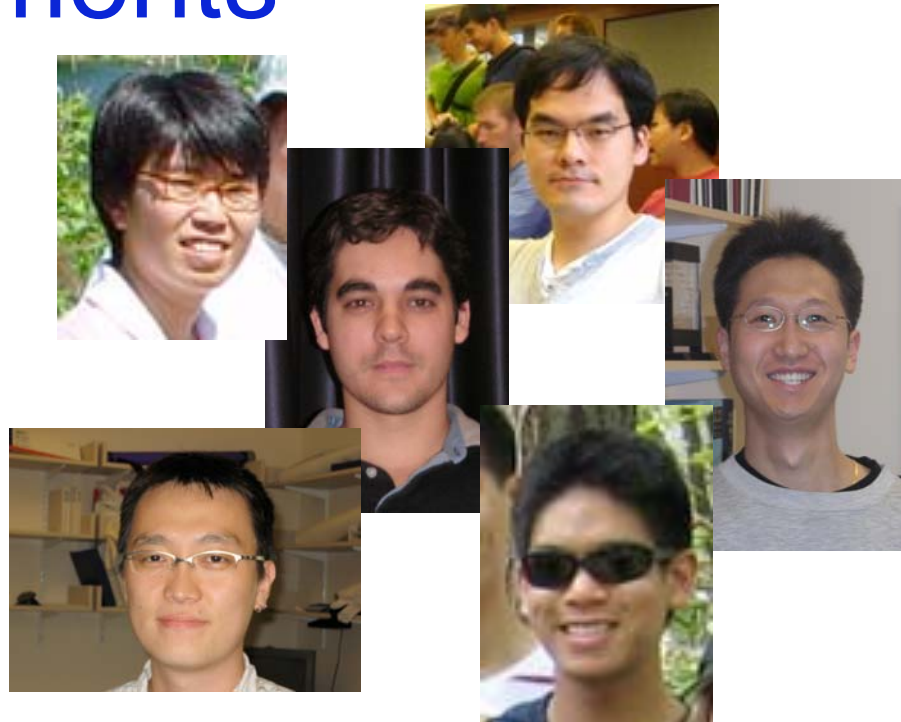
- Simulated networks exhibit structures and rheology “similar” to those found in experiments in cross-linked gels.
-
- Can a computational model be created that replicates cellular measurements?
 - What is the fundamental basis for cytoskeletal fluidization, correct power-law scaling?
 - What is the role of measurement method (1- vs. 2-particle methods, active vs. passive, bead coating, particle size vs. mesh size) in cell rheological measurements?

"If we knew what it was we were doing, it would not be called research, would it?" - Albert Einstein

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